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Growth and properties of epitaxial rare-earth scandate thin films

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ABSTRACT Epitaxial rare-earth scandate thin films of 100–1500 nm in thickness have been prepared by pulsed laser deposition on SrTiO₃(100) and MgO(100) substrates. Stoichiometry and crystallinity were investigated by Rutherford backscattering spectrometry/channelling (RBS/C), transmission electron microscopy, and X-ray diffraction. Electrical measurements on microstructured capacitors with a SrRuO₃ bottom electrode and Au top contacts reveal dielectric constants of 20 to 27, leakage currents of 0.85 to 6 $\mu\text{A}/\text{cm}^2$ at 250 kV/cm, and breakdown fields of 0.6 to 1.2 MV/cm. The optical bandgaps of the films range from 5.5 to 6 eV. The results substantiate the high potential of rare-earth scandates as alternative gate oxides.

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1 Introduction

The international technology roadmap for semiconductors (ITRS) emphasises the need for alternative gate dielectrics for silicon-based metal-oxide-semiconductor field-effect transistor (MOSFET) devices [1]. Accordingly, new high dielectric constant (κ) gate dielectrics have gained considerable attention. High- κ dielectrics are expected to provide comparable performance to SiO₂, while simultaneously permitting a thicker dielectric layer to dramatically reduce charge tunnelling through the oxide. Consequently for any alternative dielectric a high κ and a wide bandgap to limit current leakage are of primary importance.

The rare-earth scandates (REScO₃, RE denoting a rare earth element) in amorphous or epitaxial form have been proposed as candidates for the replacement of SiO₂ [2]. Their structure is orthorhombic at room temperature, isostructural with GdFeO₃ (and NdGaO₃, SrRuO₃, etc.). For example, the lattice parameters of GdScO₃ are $a = 548.8$ pm, $b = 574.6$ pm, and $c = 793.4$ pm; the space group is $Pbnm$ [3, 4]. These lattice parameters are favourable for epitaxial growth on silicon. Furthermore, all rare-earth scandates except EuScO₃ [5] are predicted to be thermodynamically stable in

contact with silicon [2]. Since there is no data available for LaScO₃ single crystals, possibly due to its high melting temperature (2290 °C, [4]), the properties reported in this work on epitaxial films are of particular value. Furthermore, initial results on thin amorphous scandate films also indicate properties advantageous for application [6, 7].

2 Sample preparation

Epitaxial and amorphous films of LaScO₃, GdScO₃, and DyScO₃ were grown by pulsed laser deposition (PLD), using a KrF excimer laser (wavelength 248 nm, pulse width 20 ns and a fluence of 2.5 J/cm²). Targets were made by mixing 99.99% pure La₂O₃, Gd₂O₃, Dy₂O₃, and Sc₂O₃ powders, ball milling in isopropanol for 24 h, calcining at 1300 °C for 24 h, pressing into 25 mm diameter targets, and finally sintering at 1500 °C for 12 h. Thin films were deposited in four different PLD chambers utilizing on-axis or off-axis geometries [8–11].

In the on-axis geometry, the substrate is located in the center of the PLD plasma plume. This permits a high deposition rate with a limited area of uniformity (in this case 1×1 cm²). The substrates are positioned directly onto a SiC resistive heater. An oxygen gas partial pressure of 2×10^{-1} Pa is maintained during the deposition process.

In the off-axis geometry, the substrate surface is oriented parallel to the expansion direction of the plume. This leads to a lower deposition rate, but allows much larger substrates (e.g., up to 2 inch diameter) to be coated uniformly. Radiation heating is provided to both sides of the substrate. Both oxygen and nitrogen gas are used as the process gas at a partial pressure of 1.2×10^{-1} Pa. Nitrogen is admitted in order to investigate the growth of scandates in a non-oxidizing atmosphere [12]. In all setups used, the prepared thin films showed comparable properties. Although the PLD process tends to generate small particulates in the films (droplets or larger particles removed from the target surface), this imposes no restriction because a sufficient area of the sample is unaffected.

Different sets of films were fabricated. LaScO₃, GdScO₃, and DyScO₃ films were deposited in an on-axis geometry on SrTiO₃(100) substrates at temperatures from 500 °C to 900 °C to establish the crystalline quality as a function of growth temperature. For optical characterization, epitaxial and amorphous LaScO₃, GdScO₃, and DyScO₃ films were

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grown in on-axis geometry on MgO(100) substrates ($1 \times 1 \text{ cm}^2$) with a 10 nm thick BaTiO₃ interlayer (necessary for epitaxial growth on MgO) also grown with PLD. For the epitaxial LaScO₃ and GdScO₃ samples on MgO, a substrate temperature of 750 °C was used, and for the epitaxial DyScO₃ 850 °C was used during deposition.

For electrical characterization, films were deposited in on-axis and off-axis geometry onto an epitaxial SrRuO₃ quasi-metallic bottom electrode on SrTiO₃(100) substrates of $1 \times 1 \text{ cm}^2$ size. The SrRuO₃ electrode was also grown by PLD at a temperature of 600 °C in a 15 Pa oxygen ambient. The substrate temperature during scandate film growth was around 1000 °C in off-axis PLD and around 850 °C in on-axis PLD. Finally, circular Au metal top electrodes were deposited by thermal evaporation through a shadow mask to form an array of parallel plate capacitors, with areas ranging from 0.00625 to 0.8 mm².

3 Film characterization

Rutherford backscattering spectrometry/channelling (RBS/C) utilizing He⁺ ions with an energy of 1.4 MeV was used to investigate the composition and crystalline quality of the films. The computer software RUMP provided the numerical analysis of the RBS data [13]. Figure 1 shows the RBS/C spectra of a 350 nm thick epitaxial LaScO₃ film grown on SrTiO₃ at 900 °C. The channelling minimum yield χ_{\min} is 2%, revealing good crystallinity. RBS analysis of a 1.5 μm thick GdScO₃ film on a BaTiO₃-buffered MgO substrate shows a Gd/Sc ratio of 1.0/1.07 and $\chi_{\min} = 4\%$, indicating that the films are stoichiometric within experimental error and have good epitaxial quality. Similar results were obtained on BaTiO₃-buffered MgO substrates for 1.5 μm thick LaScO₃ (La/Sc ratio: 1.0/0.97, $\chi_{\min} = 3\%$) and DyScO₃ (Dy/Sc: 1.0/0.95, $\chi_{\min} = 4.5\%$).

The RBS/C spectra of GdScO₃ deposited onto SrRuO₃/SrTiO₃ in the on-axis setup show a minimum yield $\chi_{\min} = 6.2\%$, demonstrating epitaxial growth. A better minimum yield of $\chi_{\min} = 3.1\%$ was achieved for LaScO₃ on the same electrode/substrate system. For comparison, a 160 nm thick GdScO₃ film on SrRuO₃/SrTiO₃ deposited in the off-axis setup exhibits an excellent Gd/Sc ratio of 1.0/1.0 as well. The use of nitrogen as a non-oxidizing process gas was tested successfully, because no indication of nitrogen was found in the films by RBS. Similar results were obtained for LaScO₃ and DyScO₃ films.

The lattice parameters of the films and their orientation were characterized by X-ray diffraction (XRD). An example is shown in Fig. 2 for LaScO₃ grown at different substrate temperatures. At 500 °C, only the XRD reflections from the SrTiO₃ substrate are detected, whereas at higher growth temperatures (600 °C and up), the XRD peaks from the scandate layer appear.

Four-circle XRD and TEM (described below) indicate that the scandate films crystallized with orthorhombic structure as found in bulk material [3, 4]. The rocking curve full width at half maximum (FWHM) of the LaScO₃ 220-reflection of the 900 °C sample is $\Delta\omega \approx 0.1^\circ$, also indicating good crystallinity. The lattice parameters of this film were determined to be $a, b = 584 \text{ pm}, c = 806 \text{ pm}$ using a Nelson–Riley

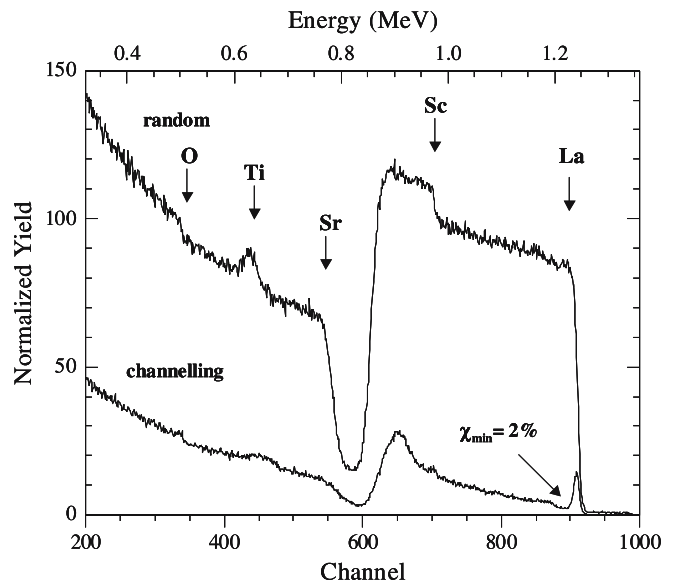


FIGURE 1 RBS random and channelling spectrum of a 350 nm thick LaScO₃ film grown epitaxially on SrTiO₃ at a substrate temperature of 900 °C. The incident energy is 1.4 MeV, the scattering angle is 170°

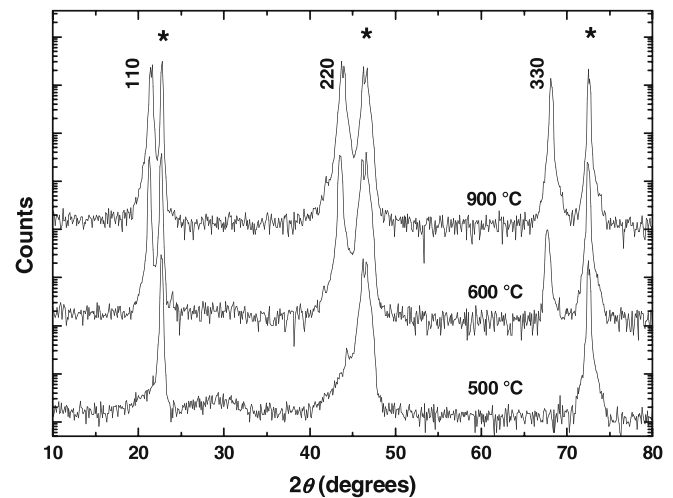


FIGURE 2 XRD θ - 2θ -scan of LaScO₃ on SrTiO₃(100) grown at different substrate temperatures. Substrate peaks are marked with an asterisk (*). The film grown at 900 °C (whose RBS spectra are shown in Fig. 1) has lattice constants $a, b = 584 \text{ pm}$ and $c = 806 \text{ pm}$, and a rocking curve FWHM of the 220 peak of 0.1°

plot [14]. Similar results were obtained for DyScO₃ (the epitaxial growth starts at 900 °C, lattice parameters $a, b = 567 \text{ pm}$ and GdScO₃ films (epitaxial growth at $> 700^\circ\text{C}$, lattice parameters $a, b = 570 \text{ pm}, c = 788 \text{ pm}$). In φ -scans performed on all scandate thin films with (110) orientation using the 226 reflection we observe four peaks separated by 90° indicating that the film is epitaxial and consists of two orientation variants related by a 90° in-plane rotation.

This is also corroborated by transmission electron microscopy (TEM) measurements. Specimens were prepared by conventional cross-sectional TEM techniques, including mechanical grinding, polishing, and dimpling, followed by final Ar-ion milling in a Gatan precision ion polishing system at 4.2 kV to electron transparency. HRTEM studies were carried out using a JEOL 4000EX operated at 400 kV, which has

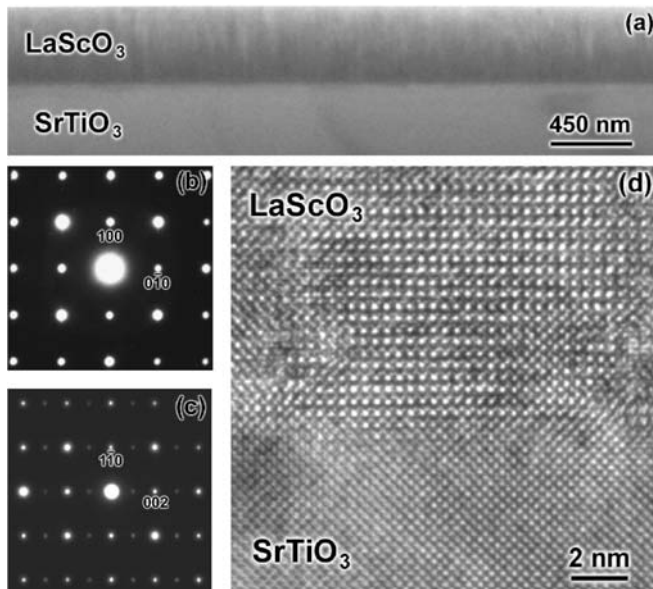


FIGURE 3 TEM characterization of the same LaScO₃ film grown on SrTiO₃(100) at 900 °C whose RBS spectra are shown in Fig. 1. (a) Low magnification image, (b) [001] zone axis SAED pattern of the SrTiO₃ substrate, (c) [110] zone axis SAED pattern of the LaScO₃ film, (d) HRTEM image of the substrate/film interface

a point-to-point resolution of 0.17 nm. Figure 3 displays the TEM results of the same LaScO₃ film whose RBS and XRD spectra are shown in Figs. 1 and 2. Figure 3a is a low magnification cross-sectional TEM image showing a very smooth surface of the film. Figure 3b and c are selected area electron diffraction (SAED) patterns taken from the substrate and the film with the incident electron beam along the same direction, respectively. They are identified as the [001] zone axis diffraction pattern of SrTiO₃ and the [110] zone axis diffraction pattern of LaScO₃, respectively. The epitaxial growth of the film is evident, and the epitaxial orientation relationship is (1 $\bar{1}$ 0) LaScO₃ || (100) SrTiO₃ with two in-plane variants: [110] LaScO₃ || [001] SrTiO₃ and [001] LaScO₃ || [001] SrTiO₃ (SAED pattern not shown). A high-resolution transmission electron microscopy (HRTEM) image of the film/substrate interface with the electron beam incident along the [001] zone axis of SrTiO₃ is shown in Fig. 3d. One can see that the interface is well defined and no noticeable interdiffusion occurs, despite the high growth temperature (900 °C).

An XRD $\theta - 2\theta$ -scan of a LaScO₃ film grown on BaTiO₃/MgO showed the lattice parameters $a, b = 578$ pm using a Nelson–Riley [14] plot of the $hh0$ peaks. The reference data of bulk LaScO₃ are $a = 568$ pm, $b = 579$ pm [3, 4]. The rocking curve FWHM of the 330 reflection is $\Delta\omega \approx 0.43^\circ$. For the GdScO₃ and DyScO₃ films grown on BaTiO₃/MgO, the lattice parameters were determined to be $a, b = 566$ pm and $a, b = 563$ pm, respectively. The bulk data of GdScO₃ are $a = 549$ pm, $b = 575$ pm and of DyScO₃ are $a = 544$ pm, $b = 571$ pm [3, 4].

4 Optical properties

Absorption measurements [15] on the scandate films on the BaTiO₃-buffered MgO revealed an optical bandgap > 5 eV. The amorphous LaScO₃ film shows an onset

of transmission at 5.5 eV, the epitaxial film at > 5.8 eV. Similar values were obtained for GdScO₃ and DyScO₃. The results are described in detail elsewhere [16]. They are comparable to photoconductivity measurements, performed on similar samples [7]. The observed bandgap is encouraging for the use as an alternative gate oxide [1, 2].

5 Electrical properties

The impedance of metal–insulator–metal parallel-plate capacitors was measured with an HP4192A impedance analyzer. The results for an epitaxial 105 nm thick DyScO₃ film (grown by on-axis PLD) are shown in Fig. 4. In the voltage range from -4 to 4 V, a 0.05 mm² electrode capacitor shows a capacitance of about 90 pF almost independent of frequency from 10 kHz to 1 MHz, corresponding to a dielectric constant κ of 21 to 22. The observed losses are low: $\tan \delta$ is in the range from 0.02 (at 10 kHz) to 0.06 (at 1 MHz). Similar results were achieved for LaScO₃ ($\kappa = 24 - 27$, $\tan \delta = 0.004 - 0.03$) and GdScO₃ ($\kappa = 20$, $\tan \delta = 0.004 - 0.03$). These values correspond well to the results measured on sin-

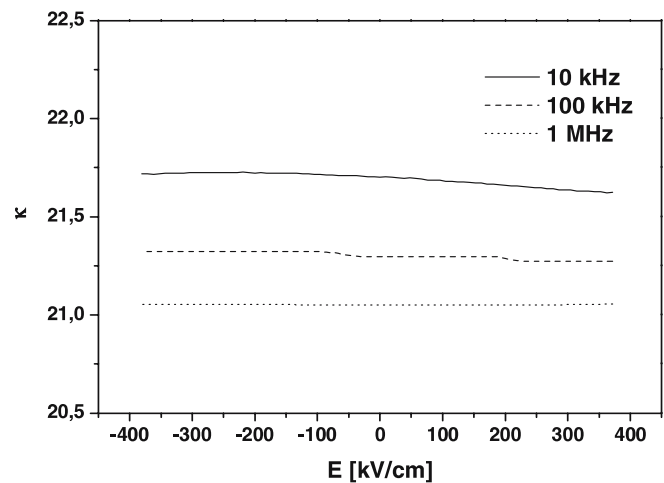


FIGURE 4 Dielectric constant of an epitaxial, 105 nm thick DyScO₃ film with SrRuO₃ bottom electrode and Au top electrode grown on a SrTiO₃(100) substrate at 850 °C

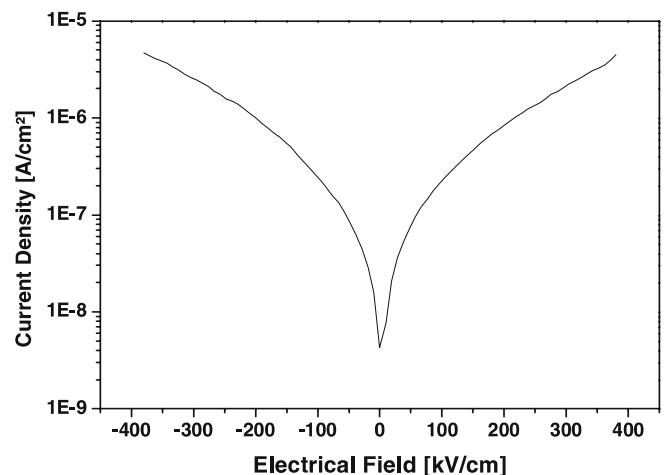


FIGURE 5 Leakage current density of the same epitaxial DyScO₃ film whose κ is shown in Fig. 4

gle crystals of GdScO₃ and DyScO₃ for (110) plates [17]. Comparable results were also observed for samples grown in a nitrogen atmosphere by off-axis PLD.

The leakage currents of the samples were investigated with an HP4155B semiconductor parameter analyzer. A representative result is shown in Fig. 5 for the 0.05 mm², 105 nm thick DyScO₃ capacitor. Leakage currents through this device are in the range of 0.67 nA at an applied voltage of 2.6 V, corresponding to a current density of 1.4 μA/cm² at an electrical field of 250 kV/cm. For GdScO₃ with a film thickness of 112 nm, the leakage current density is only 0.85 μA/cm² at 250 kV/cm, whereas a 112 nm thick LaScO₃ film shows a slightly higher leakage of 6 μA/cm² at 250 kV/cm. Dielectric breakdown of GdScO₃ occurs at 1.2 MV/cm. For DyScO₃ breakdown is observed at 1 MV/cm and for LaScO₃ it takes place at over 600 kV/cm.

6 Summary and conclusion

PLD from sintered targets has resulted in good quality, epitaxial scandate films of GdScO₃, LaScO₃, and DyScO₃. In order to study the epitaxy, several different crystalline oxides have been used as substrates. All films show relatively high dielectric constants κ in comparison to other alternative gate dielectric candidates [2]. In addition, low leakage currents were observed. Optical measurements demonstrate a wide bandgap of the amorphous (> 5.5 eV) as well as of the epitaxial (> 5.8 eV) films. Scandate films could also be grown in an oxygen-free atmosphere with correct stoichiometry and comparable electrical properties. Such an oxygen-free deposition route suggests the possibility of growing scandate films onto silicon without a parasitic SiO₂ interlayer. Finally, the results illustrate the promise of rare earth scandates as the gate insulator for future silicon-based MOSFET devices. Presently, studies on thinner layers with an equivalent oxide thickness below 1 nm are in progress.

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